## Effect of dielectric environment on the ultraviolet optical absorption of single-walled carbon nanotubes

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We studied optical absorption of single-walled carbon nanotubes by varying the dielectric environment. For the two different components of the broad UV-absorption feature conventionally referred to as the  $\pi$  plasmon, we find that the component at 5.0–5.3 eV exhibits remarkable spectral changes, which we attribute to a dipolar radial surface plasmon. However, the component at ~4.5 eV remains unchanged, raising a fundamental question as to its conventional attribution. We discuss its relation with the absorption feature at ~4.5 eV in graphite arising from an interband transition.

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Single-walled carbon nanotubes (SWNTs) are a novel one-dimensional (1D) material made of an  $sp^2$ -bonded wall one atom thick. Strong confinement (~1 nm) of charge carriers in SWNTs results in their unique optical properties being dominated by strongly bound excitons (Coulomb-bound electron-hole pairs),<sup>1–3</sup> recognized by the sharp opticalabsorption and photoluminescence peaks in near-infrared (NIR) region.<sup>4</sup> Due to this aspect, SWNTs have been proposed for various optical applications such as NIR detectors and light emitters.<sup>5,6</sup>

It is known that these sharp optical features accompany an optical-absorption baseline extending from the UV region, where there is a broad and intense optical-absorption peak in the 4–6 eV range that is typically attributed to a  $\pi$ -plasmon excitation.<sup>7–9</sup> It has been pointed out that the intensity ratio between the excitonic absorption and NIR baseline absorption increases as the degree of SWNT isolation is enhanced,<sup>10</sup> although this mechanism has not yet been clarified. So far, several theoretical studies have investigated UV absorption in SWNTs,<sup>11–14</sup> but many aspects are inconsistent and are still under debate. Therefore, clear understanding of the UV-absorption features is essential to fully utilize SWNTs for suggested optical applications.

Previously, we showed that the broad UV-absorption peak in the 4-6 eV range actually consists of two distinct components.<sup>15,16</sup> In particular, the absorption feature at  $\sim$ 4.5 eV is observed for light polarized parallel to the SWNT axis, while the other absorption feature at 5.0-5.3 eV is observed for light polarized perpendicular to the axis.<sup>15</sup> These polarization dependences have recently been confirmed by optical-absorption measurements of highly individualized SWNTs (Refs. 17 and 18) and angle-resolved electron-energy-loss spectroscopy (AR-EELS) of aligned SWNTs.<sup>19</sup> Also in our previous report, we attributed the parallel component ( $\sim$ 4.5 eV) to the maximum of the imaginary part of the dielectric function perpendicular to the graphene c axis (i.e., in-plane direction)  $Im\{\varepsilon_{\perp}\}$ , and the perpendicular component (5.0–5.3 eV) to the maximum of Im{  $-\varepsilon_{\parallel}^{-1}$  parallel to the *c* axis (out-of-plane direction).<sup>15</sup>

In this paper, we show that the UV-absorption feature in the 5.0-5.3 eV range is sensitive to changes in the surrounding dielectric environment, while the other absorption feature at  $\sim$ 4.5 eV is unaffected. Based on these experimental results, we discuss the properties as well as physical origins of the two UV-absorption features.

We used vertically aligned SWNTs (VA-SWNTs) directly grown on a quartz substrate<sup>20,21</sup> by the alcohol chemicalvapor deposition method<sup>22</sup> for the optical-absorption measurements. Figure 1(a) shows a field-emission scanning electron microscope (FE-SEM) image of the sample. The thickness of the VA-SWNT film is typically  $5-10 \mu$ m. The order parameter is known to be ~0.75,<sup>15,16</sup> which corresponds to an average deviation of 24° from the ideal alignment. The SWNTs form thin bundles with diameters  $\leq 10$  nm (Ref. 23). Figure 1(b) shows a high-resolution transmission electron microscopy (HR-TEM) image of VA-SWNTs transferred directly onto a transmission electron microscopy (TEM) grid by a mechanical (dry) process. The

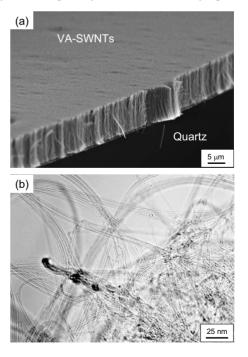


FIG. 1. (a) FE-SEM image of VA-SWNTs grown on the surface of a quartz substrate. (b) HR-TEM image of VA-SWNTs directly transferred onto a TEM grid.

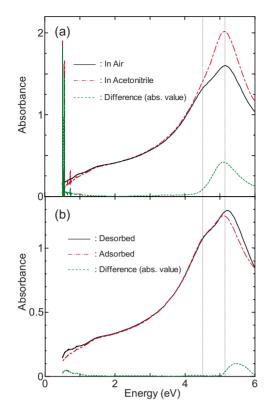


FIG. 2. (Color online) (a) Comparison of the optical-absorption spectra of the same VA-SWNT sample measured in air (black solid) and subsequently in acetonitrile (red dot dash). Spikes below 0.8 eV are absorptions by acetonitrile. (b) Comparison of the optical-absorption spectra of an identical VA-SWNT sample measured in a molecular-adsorbed state (red dot dash) and after desorption of the molecules (black solid). In panels (a) and (b), green dashed curves show absolute values of the differences between the two spectra and vertical dotted lines indicate the positions of the two distinct UV-absorption features. All spectra are shown *as measured*, i.e., no scaling operations have been performed.

image shows that the SWNTs are free from amorphous carbons and graphitic impurities. The high quality and purity of our VA-SWNTs have been confirmed by the high "G-to-D ratio" from Raman scattering measurements<sup>20</sup> and by the result of an "optical thermogravimetric analysis (TGA)" measurement where VA-SWNTs burned only at around 600 °C (Fig. 8 of Ref. 24). In the following, optical-absorption spectra of VA-SWNTs were measured with light normally incident on the substrates, i.e., polarized perpendicular to the direction of ideal alignment of the VA-SWNTs.

Figure 2(a) compares two absorption spectra, the spectrum of the sample measured in air (solid curve) and that of the same sample subsequently measured in acetonitrile (dot-dash curve). Acetonitrile (CH<sub>3</sub>CN) has a high dielectric constant ( $\varepsilon \sim 38$ ) and high optical transparency up to 6 eV. Both of these measurements were performed by placing the sample inside a quartz cuvette, whose optical density was negligible for the entire energy range of the measurement. The optical absorption in NIR was reduced by the immersion of the sample into acetonitrile and is explained by the change in the Fermi level caused by the charge transfer between SWNTs and the surrounding medium.<sup>25,26</sup> At higher

energies, the magnitude of this difference becomes smaller until both spectra coincide for photon energies above 1.5 eV. However, the spectral change is again observed in the UV region around 5 eV, which was not known previously.

Figure 2(b) compares spectra measured from a VA-SWNT sample in a "desorbed" state (minimal molecular adsorption, black-solid curve) and an "adsorbed" state (significant molecular adsorption, red dot-dash curve). The adsorbed state was achieved by storing the sample in a container filled with air for four months after its synthesis. The desorbed state was achieved by heating the same sample for 1 h at 200 °C in a low-pressure Ar atmosphere to desorb molecules from the SWNTs. The measurement of the adsorbed state was performed first, then heat treatment, which was immediately followed by the measurement of the same desorbed sample. Figure 2(b) also shows reduction in the absorption intensity in NIR. Exposure of SWNTs to air is known to result in *p*-type doping caused by adsorption of molecular oxygen and/or O-H species.<sup>27,28</sup> Similar to the case of acetonitrile immersion [Fig. 2(a)], a spectral change is also seen in UV. Figures 2(a) and 2(b) clearly show that only the opticalabsorption feature at 5.0-5.3 eV exhibits a spectral change, while both features at  $\sim 4.5$  eV and the absorption in the intermediate region (2-4 eV) do not.

We show that the same optical change can be induced for the 5.0-5.3 eV component by a simple electrochemical method. The measurement setup consists of a VA-SWNT substrate [working electrode (WE)], a Pt wire [counter electrode (CE)], and an Ag wire [reference electrode (RE)], set in an optically transparent quartz cell filled with acetonitrile and 0.1 M lithium perchlorate (LiClO<sub>4</sub>), which served as an electrolyte. The voltage was applied between the WE and CE, while the magnitude of the applied voltage was measured between the WE and RE. Figure 3(a) shows the change in optical absorption as the voltage was varied from 0.25 to 1.0 V. Figure 3(b) shows the subsequent reverse process (1.0-0.25 V). In addition to the expected reduction in optical absorption below 2 eV, we also see a change in the magnitude of the 5.0-5.3 eV component that has been enhanced due to the immersion in acetonitrile [as seen in Fig. 2(a)]. Figure 3(c) plots the induced changes at 1.3 and 5.1 eV. Each measurement was taken approximately 5 min after each voltage change. For 5.1 eV, the last measurement at 0.25 V (marked with an asterisk) was taken 15 min after the previous measurement at the same voltage, indicating a much slower response for the optical change at 5.1 eV than that at 1.3 eV.

As shown in Figs. 2 and 3, the perpendicular component (5.0–5.3 eV) is influenced by changes in the environment. In the case of 1D metallic nanowires, it is known that the direction perpendicular to the nanowire axis (the direction of electron confinement) is the only direction in which dipolar surface plasmons can be excited.<sup>29,30</sup> Figure 4 shows schematics of dipolar surface plasmons induced on the surface of cylindrical nanowire (viewed along the axis direction) with complex dielectric function  $\varepsilon$ . The applied external field and dielectric function of the surrounding medium are denoted by  $E_0$  and  $\varepsilon_m$ , respectively. The depolarization field  $E_{dep}$  created by induced charges on the interface is given by<sup>29</sup>

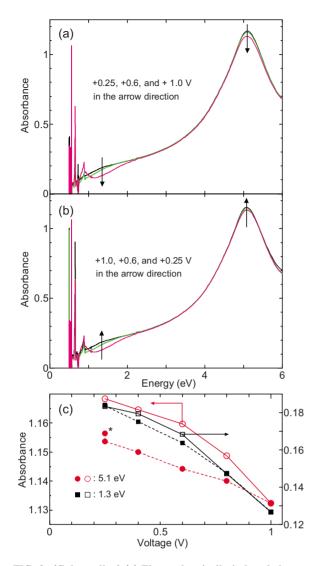


FIG. 3. (Color online) (a) Electrochemically induced changes in the absorption spectra measured in 0.1 M LiClO<sub>4</sub> in acetonitrile. (b) The voltage was changed from 0.25 to 1.0 V [panel (a)] and subsequently from 1.0 to 0.25 V [panel (b)]. (c) Absorbance changes measured at 1.3 (black squares) and 5.1 eV (red circles). Open symbols connected by solid lines are for the  $0.25 \rightarrow 1.0$  V process, while filled symbols connected by dotted lines are for 1.0  $\rightarrow 0.25$  V. The last point for 5.1 eV at 0.25 V (marked by an asterisk) was measured 15 min after the previous measurement at the same voltage. Three spectra shown in panels (a) and (b) are from the experiment in panel (c), corresponding to 0.25, 0.6, and 1.0 V.

$$E_{\rm dep} = -\frac{\varepsilon - \varepsilon_m}{\varepsilon + \varepsilon_m} E_0. \tag{1}$$

The resultant field inside the nanowire  $E_{in}$  proportional to optical-absorption intensity is

$$\boldsymbol{E}_{\rm in} = \boldsymbol{E}_0 + \boldsymbol{E}_{\rm dep} = \frac{2\boldsymbol{\varepsilon}_m}{\boldsymbol{\varepsilon} + \boldsymbol{\varepsilon}_m} \boldsymbol{E}_0, \qquad (2)$$

which is sensitive to the value of  $\varepsilon_m$ . Furthermore, the photon energy corresponding to the maximum dipolar surface-

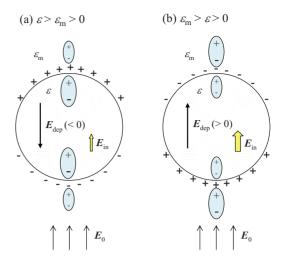


FIG. 4. (Color online) Schematics of dipolar surface plasmons induced on the surface of cylindrical nanowires by an external electric field  $E_0$  viewed along the axial direction.  $\varepsilon$ ,  $\varepsilon_m$ : complex dielectric functions of the nanowire and surrounding media, respectively.  $E_{dep}$ ,  $E_{in}$ : depolarization electric field and the resultant field inside the nanowire, respectively. Panels (a) and (b) depict  $\varepsilon_m > \varepsilon$ >0 and  $\varepsilon > \varepsilon_m > 0$  cases, respectively.

plasmon excitation depends on both  $\varepsilon_m$  and the charge density (or plasma frequency ) of the material.<sup>29</sup> Although a very strong "resonance" condition can occur when  $\varepsilon \cong -\varepsilon_m$ [where the denominator of Eq. (2) is diminished], we consider here only  $\varepsilon$ ,  $\varepsilon_m > 0$  cases shown in Figs. 4(a) and 4(b).

Changes observed in the perpendicular component are well explained in terms of dipolar surface plasmons. The significant enhancement of this absorption [Fig. 2(a)] is explained by the much larger  $\varepsilon_m$  of acetonitrile than of air [Eq. (2)]. The weakening and the redshift of this feature observed in Fig. 2(b) may be ascribed to *p*-type doping (withdrawal of electrons from SWNTs) caused by the adsorption of oxygen molecules,<sup>27,28</sup> as well as a change in  $\varepsilon_m$  due to the adsorbed molecules. More importantly, the difference in the observed transient changes in the absorbance at 1.3 and 5.1 eV [Fig. 3(c) suggests that the dominant mechanisms responsible for the optical changes at these energies are different. The former is caused by the change in the Fermi level, which always results in the reduction in absorption intensities regardless of p- or *n*-type doping.<sup>25</sup> On the other hand, the much slower change observed at 5.1 eV suggests that this feature is influenced by a slower process such as diffusion of the electrolyte around the SWNTs. This is plausible considering that 5.1 eV is considerably far from the Fermi level and no optical changes were observed in the intermediate (2-4 eV) region.

On the other hand, no change was observed at ~4.5 eV (parallel component) by the change in the environment as shown by Figs. 2(a) and 2(b). It is noted that this feature is, at least, not a dipolar surface plasmon because this direction is not the direction of confinement. So far, the parallel component at ~4.5 eV has been regarded as a  $\pi$  plasmon.<sup>7–9</sup> However, in the case of SWNTs *all atoms constitute the surface*, thus the  $\pi$  electrons are directly exposed to the surrounding medium. If this feature truly originated from a  $\pi$ 

plasmon, it is expected to be affected by the change in dielectric environment. Since this is not observed, our experimental results raise a fundamental question as to the physical origin of this parallel component of UV absorption. A recent AR-EELS study of the same VA-SWNTs has also recognized the peak component at ~4.5 eV by extrapolating the obtained electron-energy-loss spectroscopy (EELS) spectra to the case of zero-momentum transfer ( $q \rightarrow 0$ , analogous to optical excitation).<sup>19</sup> However, we note that the observation of a peak in EELS is not sufficient to conclude that the peak arises from plasmons because *interband electron transitions can also be excited in EELS* (although as an indirect transition accompanying nonzero q).<sup>31</sup>

In the case of graphite, the origin of the strong UVabsorption peak at ~4.5 eV has been recognized as a  $\pi \rightarrow \pi^*$  interband electronic transition at the *M* (also called *Q*) point of the Brillouin zone, corresponding to the maximum of Im{ $\varepsilon_{\perp}$ }.<sup>32–35</sup> One of the most essential differences between SWNTs and graphite/graphene is their dimension. In a pure 1D system, it is known that only collective excitations can exist.<sup>36</sup> Based on this viewpoint, electronic excitations in SWNTs are essentially collective as long as SWNTs are regarded as a 1D system. In this context, we must address the issue of terminology, i.e., what to call the UV-absorption feature at ~4.5 eV in SWNTs. However, apart from semantics, it is more important to clarify its physical origin. Further theoretical investigations are required to conclude the issues discussed above, especially regarding the origin of the parallel UV-absorption feature of SWNTs observed at  $\sim$ 4.5 eV.

In summary, we studied the two UV-absorption features of SWNTs in the 4–6 eV range that have so far been referred to as  $\pi$  plasmons. We found that changes in the dielectric environment induce remarkable changes in the perpendicular component at 5.0–5.3 eV. This feature is explained as a radial dipolar surface plasmon induced in SWNTs. On the other hand, we found that the parallel UV-absorption component at ~4.5 eV was unaffected by the environmental changes. This raises a fundamental question as to the physical origin of this absorption feature, which has heretofore been classified as a  $\pi$  plasmon. We anticipate that the findings and discussions presented in this paper will stimulate further investigations and elucidation of the properties of UV absorption in SWNTs.

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